

# Geometry, electronic structure, and magnetic ordering of iron–carbon nanoparticles

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**Abstract** The geometry optimization of  $\text{Fe}_n\text{C}_m$  and  $\text{Fe}_n\text{C}_m^+$  nanoparticles from  $\text{FeC}_4$  to  $\text{Fe}_7\text{C}_8$  was carried out using DMol<sup>3</sup> method. The most stable isomers for neutral and charged clusters are found with the use of ‘binomial’ scheme. For each investigated composition, the value of binding energy per atom for the ground isomer was evaluated as well as the number of configurations with binding energies which are close to that of the ground geometry. In almost all compositions, the isomers with ferromagnetic ordering of spins on Fe atoms are found to have the greater dissociation energy than those with any other spin ordering.

**Keywords**  $\text{Fe}_n\text{C}_m$ : ground isomers · Relative stability · Magnetic ordering

## 1 Introduction

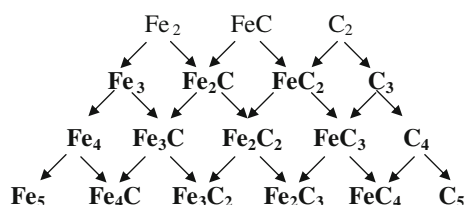
Since the discovery of the metallocarbohedrene  $\text{Ti}_8\text{C}_{12}$  in 1992 [1], the molecular systems  $\text{M}_x\text{C}_y$  became the subjects of numerous experimental and theoretical investigations. The electronic structure of nanoparticles containing iron atoms  $\text{Fe}_n\text{C}_m$  is also interesting due to the position of Fe between the early and late transition metals (TM). Theoretical investigations of  $\text{M}_x\text{C}_y$  clusters containing the early (V) and late TM (Ni, Co) were presented in [2–5]. There are two ways to generate small Fe–C clusters. Firstly, such

objects can be produced by grinding of solid-phase carbides. Though the maximum carbon content in all crystalline Fe carbides  $\text{Fe}_x\text{C}_y$  does not exceed 1/3 ( $y/x \leq 1/3$ ) [6], the  $\text{Fe}_n\text{C}_m$  fragments of crystal lattice could be of various stoichiometry when the object dimensions are small ( $m, n < 20$ ). Second way of binary  $\text{Fe}_n\text{C}_m$  clusters generation is the laser vaporization of Fe metal and cooling in acetylene. In experimental work of Pilgrim and Duncan [7], the distribution of cationic  $\text{Fe}_n\text{C}_m^+$  particles had been obtained. According to these results, the first observed cluster has (5,6) composition and small relative abundance. The most abundant  $\text{Fe}_n\text{C}_m^+$  species have (7,8), (8,12), and (12,12) content [7]. Theoretical modeling of iron–carbon clusters also follows two ways: Harris and Dance [8] started from  $\text{Fe}_{12}\text{C}_{12}$  fragment of the cubic crystal lattice and investigated the dissociation series  $\text{Fe}_{12}\text{C}_{12} \rightarrow \text{Fe}_2\text{C}_{12}$ . However, the more popular way to construct the initial configurations for geometry optimization was the use of combinations of iron and carbon dimers or  $\text{C}_n$  chains or the analogs of known metallocarbohedrenes [9–16].

The serious difficulty in theoretical investigations of various  $\text{Fe}_n\text{C}_m$  clusters is the existence of a large amount of isomers for each composition ( $n, m$ ) and the absence of a fast scheme for reliable generation of the ground geometrical structure. Usually, the search for stable geometries results in the definition of some part of the possible isomers, the lowest energy configuration is considered as the ground geometry for this  $\text{Fe}_n\text{C}_m$  particle. Earlier in our investigations of electronic structure of small iron–carbon clusters [17, 18], the ‘binomial’ scheme for the generation of stable isomers was suggested (Fig. 1). From our point of view, this scheme is a reliable way to find all possible geometries for any composition ( $n, m$ ). The shortcoming of this method is the fast increasing amount of isomers as the number of atoms increases, especially for the cases  $n/m \sim 1$

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**Fig. 1** The scheme for search of the stable structures for  $\text{Fe}_n\text{C}_m$  particles

[18]. However, the analysis of the parentage of the ground configurations for  $\text{FeC}_2$ ,  $\text{Fe}_2\text{C}$ ,  $\text{FeC}_3$ ,  $\text{Fe}_3\text{C}$ ,  $\text{Fe}_2\text{C}_2$ , and  $\text{Fe}_2\text{C}_3$  showed that in all cases, the most stable isomer  $\text{Fe}_n\text{C}_m$  was obtained from the most stable configurations of both ‘preceding’ particles  $\text{Fe}_{n-1}\text{C}_m$  and  $\text{Fe}_n\text{C}_{m-1}$  by the addition of Fe or C atoms, respectively. Thus, to find the ground geometry for any  $\text{Fe}_n\text{C}_m$  cluster, one needs to consider only the ground isomers of all previous compositions.

The aims of the present work are: (1) the search for the probable ground structures of the  $\text{Fe}_n\text{C}_m$  and  $\text{Fe}_n\text{C}_m^+$  clusters for  $5 \leq n + m \leq 15$ ; (2) the evaluation of the trends in binding energy variation along  $n$  and  $m$  series; (3) the check-up of assumption that ferromagnetic ordering of spin moments on metal sites is preferable for the ground isomers; (4) the comparison of stabilities of the clusters generated by addition of atoms ‘one by one’ and by grinding of the most stable solid-phase carbide  $\text{Fe}_3\text{C}$ .

## 2 Objects and methods of calculations

In the present work, the search for  $\text{Fe}_n\text{C}_m$  stable structures followed the ‘binomial’ scheme, i.e., Fe and C atom approached the known ground isomers of  $\text{Fe}_2\text{C}_2$ ,  $\text{Fe}_3\text{C}$ ,  $\text{FeC}_3$ , and  $\text{Fe}_2\text{C}_3$  [17, 18] from all possible directions. Then, the resulting ground configurations for  $\text{FeC}_4$ ,  $\text{Fe}_2\text{C}_4$ ,  $\text{Fe}_3\text{C}_2$ , and  $\text{Fe}_3\text{C}_3$  served as initial objects for generating the structures of  $\text{Fe}_n\text{C}_m$  particles with greater  $n$  and  $m$  values. We did not take into account the ‘chemical preference’ of some directions because when adding the extra atom on all geometrically possible sites for forming a bond, the DMol<sup>3</sup> procedure can always ‘catch’ any nearby configuration with local minimum of energy. The two paths of each cluster generation (Fig. 1) give one a possibility for the verification of the results.

Geometry optimization of  $\text{Fe}_n\text{C}_m$  particles was performed using the DMol<sup>3</sup> program [19] in spin-unrestricted approach and with double numerical atomic basis set with d-polarization function for C (‘dnd’). The Coulombic potential was computed with the use of model density obtained as decomposition of charge density into multipolar components including those with  $l = 3$ . The



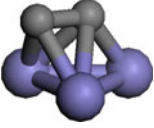
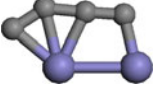
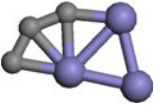

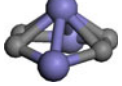
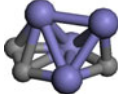
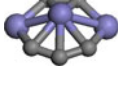
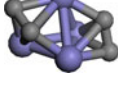
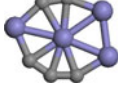

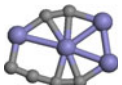
generalized gradient approximation (GGA) in ‘PBE’ [20] and ‘BLYP’ [21, 22] forms was used in all calculations. Optimization of the molecular structures was done until change in a value of maximum energy gradients was less than 0.001 atomic units, the global orbital cutoff was 8.0 Å. The thermal smearing of 0.005 was employed to accelerate the convergence of SCF process. However, the thermal smearing could lead to some increase in molecular size and dissociation energies [18]. To confirm that the stationary points correspond to minima, the vibrational frequencies were computed for all isomers obtained. The binding energy of each resulting geometry obtained from different starting configurations can vary within 0.1 eV, mainly for less symmetrical structures.

Since the DMol<sup>3</sup> code incorporates only the qualitative schemes of Mulliken [23] and Hirshfeld [24] for the calculation of the effective charges on atoms ( $Q_{\text{eff}}$ ), we also used the Discrete Variational method (DVM) [25, 26]. For each final configuration obtained by DMol<sup>3</sup>, the DV calculation was carried out, where  $Q_{\text{eff}}$  is computed as the integral of electron density inside the domain bounded by the points of its minimum [27].

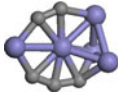
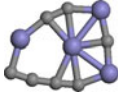
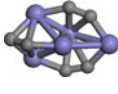
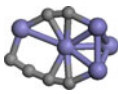
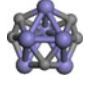
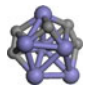
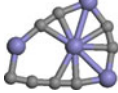
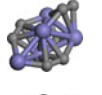
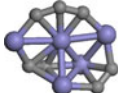
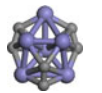
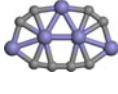
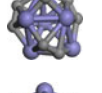
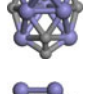
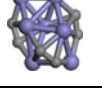
## 3 Results of geometry optimization

In Table 1, we present the results of our calculations for the neutral  $\text{Fe}_n\text{C}_m$  clusters generated according to ‘binomial’ scheme. In Appendix, for these most stable isomers, we report their atomic coordinates that could be helpful for the understanding of geometrical structures and could be used in the further theoretical and experimental researches. As can be seen, the ‘PBE’ functional noticeably overestimates the absolute values of  $E_b$  in comparison with the energies obtained using ‘BLYP’ approach. However, the difference in atomic coordinates obtained using two functionals did not exceed 0.05 Å. According to our previous results [17], the smaller values for  $|E_b|$  which can be achieved by some modification of computational scheme are closer to the measured dissociation energies than are those obtained in the spin-unrestricted DMol calculations using ‘PBE’ and ‘P91’ functionals. Noya et al. [15] obtained the same ground geometry for  $\text{FeC}_4$  cluster using Siesta program and ‘PBE’ functional; however, their  $|E_b| = 23.4$  eV is noticeably less than our ‘PBE’ (26.4 eV) and even ‘BLYP’ (24.7 eV) values. The part of this difference is due to systematic overestimation of  $-E_b$  obtained in DMol calculations for  $\text{Fe}_n\text{C}_m$  particles as compared with Siesta method, and another part of the difference is due to the thermal smearing which is used in the present calculations. On the other hand, Ma et al. [16] predicted the linear ground structure for  $\text{FeC}_4$  cluster using the same DMol method and almost the same computational procedure with

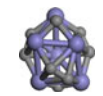
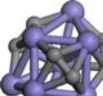
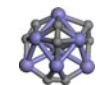
**Table 1** Geometry, binding energy (eV), energy gap (eV), and ionization energies  $E_{i1}$  and  $E_{i2}$  (eV) of the ground isomers of  $\text{Fe}_n\text{C}_m$  nanoparticles (large spheres correspond to Fe atoms and small spheres correspond to C atoms)

Composition	Shape	Symmetry	Binding energy		Energy gap		$E_{i1}$ $E_{i2}$	
			PBE	BLYP	PBE	BLYP	PBE	BLYP
$\text{FeC}_4$		$\text{C}_{2v}$	−26.4	−24.7	0.83	0.68	9.02	9.04
							8.92	8.79
$\text{Fe}_3\text{C}_2$		$\text{C}_{2v}$	−20.4	−19.0	0.71	0.72	6.14	6.09
							6.13	6.05
$\text{Fe}_3\text{C}_2$		$\text{C}_s$	−20.6	−18.8	0.53	0.47	6.44	6.32
							6.33	6.21
$\text{Fe}_2\text{C}_4$		$\text{C}_s$	−30.3	−28.5	0.82	0.84	7.80	7.41
							6.97	6.75
$\text{Fe}_3\text{C}_3$		$\text{C}_s$	−27.1	−25.2	0.83	0.80	7.01	6.93
							6.96	6.87
$\text{Fe}_3\text{C}_3$		$\text{C}_1$	−27.2	−25.1	0.63	0.60	7.23	6.99
							6.72	6.64
$\text{Fe}_3\text{C}_4$		$\text{C}_{2v}$	−34.9	−32.4	0.72	0.68	7.11	7.00
							6.80	6.67
$\text{Fe}_4\text{C}_3$		$\text{C}_1$	−31.7	−29.0	0.57	0.46	6.99	6.84
							6.95	6.79
$\text{Fe}_3\text{C}_5$		$\text{C}_s$	−42.1	−39.2	0.71	0.81	7.09	7.00
							6.99	6.91
$\text{Fe}_4\text{C}_4$		$\text{C}_2$	−38.8	−35.6	0.33	0.28	6.97	6.83
							6.92	6.79
$\text{Fe}_4\text{C}_5$		$\text{C}_s$	−46.3	−42.9	0.69	0.66	6.79	6.62
							6.75	6.57
$\text{Fe}_5\text{C}_4$		$\text{C}_{2v}$	−43.4	−39.6	0.27	0.30	6.35	6.19
							6.29	6.13
$\text{Fe}_4\text{C}_6$		$\text{C}_s$	−53.2	−49.6	0.29	0.38	7.04	6.88
							6.97	6.80

**Table 1** continued

Composition	Shape	Symmetry	Binding energy		Energy gap		$E_{i1}$ $E_{i2}$	
			PBE	BLYP	PBE	BLYP	PBE	BLYP
Fe <sub>5</sub> C <sub>5</sub>		C <sub>s</sub>	−50.1	−46.1	0.07	0.06	6.26	6.12
							6.23	6.09
Fe <sub>4</sub> C <sub>7</sub>		C <sub>s</sub>	−60.4	−56.5	0.56	0.51	6.88	6.75
							6.84	6.70
Fe <sub>5</sub> C <sub>6</sub>		C <sub>s</sub>	−57.5	−52.9	0.48	0.59	6.74	6.58
							6.69	6.50
		C <sub>1</sub>	−57.1	−53.0	0.51	0.52	6.72	6.56
							6.64	6.46
Fe <sub>6</sub> C <sub>5</sub>		C <sub>s</sub>	−54.5	−49.5	0.39	0.51	6.36	6.09
							6.26	6.08
		C <sub>1</sub>	−54.4	−49.5	0.19	0.01	6.55	6.36
							6.49	6.31
Fe <sub>4</sub> C <sub>8</sub>		C <sub>s</sub>	−67.2	−63.1	0.07	0.23	7.14	6.93
							7.05	6.79
Fe <sub>5</sub> C <sub>7</sub>		C <sub>s</sub>	−64.5	−59.4	0.72	0.59	7.31	7.30
							7.12	7.14
		C <sub>1</sub>	−64.4	−59.7	0.35	0.60	6.58	6.65
							6.53	6.46
Fe <sub>6</sub> C <sub>6</sub>		C <sub>s</sub>	−61.7	−56.9	0.41	0.49	6.52	7.18
							6.44	6.79
Fe <sub>5</sub> C <sub>8</sub>		C <sub>s</sub>	−72.2	−67.2	0.31	0.24	6.97	6.83
							6.82	6.64
Fe <sub>6</sub> C <sub>7</sub>		C <sub>1</sub>	−69.3	−63.5	0.37	0.03	6.70	6.92
							6.65	6.54
Fe <sub>7</sub> C <sub>6</sub>		C <sub>s</sub>	−65.6	−59.9	0.35	0.04	6.40	6.20
							6.35	6.14
		C <sub>1</sub>	−65.4	−60.1	0.02	0.33	6.30	6.80
							6.06	6.39

**Table 1** continued

Composition	Shape	Symmetry	Binding energy		Energy gap		$E_{i1}$ $E_{i2}$	
			PBE	BLYP	PBE	BLYP	PBE	BLYP
Fe <sub>6</sub> C <sub>8</sub>		C <sub>1</sub>	−76.4	−70.2	0.38	0.56	6.98 6.94	6.79 6.76
		C <sub>1</sub>	−75.8	−70.4	0.15	0.26	6.63 6.52	6.21 6.16
Fe <sub>7</sub> C <sub>7</sub>		C <sub>1</sub>	−73.9	−67.4	0.05	0.02	7.04 6.87	6.82 6.59
		C <sub>1</sub>	−73.9	−67.5	0.21	0.07	6.55 6.47	6.47 6.28
		C <sub>1</sub>	−81.4	−74.7	0.14	0.03	7.24 7.14	7.57 6.58
		C <sub>1</sub>						

BLYP functional. We also considered a linear structure with the Fe atom at one end as a possible candidate for FeC<sub>4</sub> ground geometry. In agreement with the results of [16], the binding energy of our C<sub>2v</sub> planar structure (Table 1) is 0.3 eV higher than that of the linear structure when the BLYP exchange–correlation potential is used. However, according to our calculations using PBE functional, the  $E_b$  of the planar structure is 0.4 eV lower than that of the linear isomer. In addition to the comparison of binding energies, the various shifts (0.1–0.3 Å) of Fe atom from equilibrium position were considered. A shift of Fe atom by 0.3 Å led to a transformation of the linear isomer into the arched chain, while the planar structure appeared to be more stable. The composition Fe<sub>2</sub>C<sub>4</sub> was also investigated in [16]; unfortunately, only the high-symmetrical cyclic configurations (C<sub>2h</sub>, C<sub>2v</sub>, D<sub>2h</sub>) were considered by the authors. Our procedure of successive addition of atoms ‘one by one’ allows one to obtain a wide series of high-symmetrical clusters as well as low-symmetrical ones. The comparison of  $E_b$  for several Fe<sub>2</sub>C<sub>4</sub> isomers obtained using both functionals shows that C<sub>S</sub> structure presented in Table 1 is more stable than all cyclic and linear structures considered in [16].

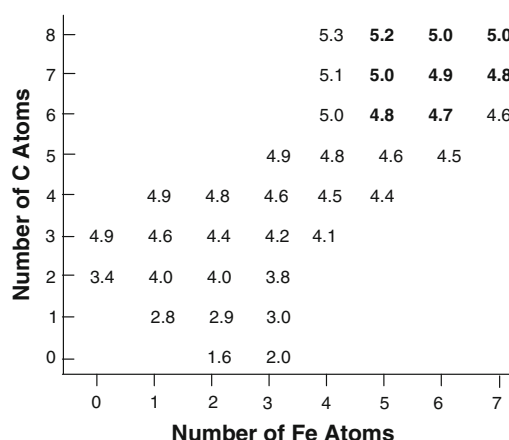
For some ( $n$ ,  $m$ ) compositions, the energetic ordering of the highest energy structures is different for ‘PBE’ and ‘BLYP’ approaches (Table 1); for Fe<sub>3</sub>C<sub>3</sub>, Fe<sub>6</sub>C<sub>5</sub>, and Fe<sub>7</sub>C<sub>7</sub>, the binding energy for two of the most stable conformations lies within 0.1 eV of each other, making it impossible to know which of them is the ground isomer. In

these cases, both of the structures were considered as the ground geometry and were used for generating the next Fe <sub>$n+1$</sub> C <sub>$m$</sub>  or Fe <sub>$n$</sub> C <sub>$m+1$</sub>  particles. Note that the highest point symmetry detected for the most stable isomers is C<sub>2v</sub> and the most frequently obtained point group is C<sub>S</sub>. The planar structures are typical for the ground isomers of small clusters with  $n + m \leq 6$  and for some particles containing four metal atoms: Fe<sub>4</sub>C<sub>5</sub>, Fe<sub>4</sub>C<sub>6</sub>, Fe<sub>4</sub>C<sub>7</sub> and Fe<sub>4</sub>C<sub>8</sub>. The latter particle is the largest planar ground structure obtained in our work, and we can suggest that Fe<sub>4</sub>C<sub>8</sub> is the ‘last’ planar ground isomer among other Fe <sub>$n$</sub> C <sub>$m$</sub>  ground geometries. However, the Fe<sub>5</sub>C<sub>8</sub> and Fe<sub>6</sub>C<sub>8</sub> (#2) clusters have the structures which are close to planar. Analysis of the shapes of ground isomers with planar configurations (or close to planar) shows that in addition to C<sub>2</sub> carbon pairs, the C<sub>3</sub> and C<sub>4</sub> carbon chains are typical for such type of geometry. In the cases of three-dimensional shape, the most common feature obtained in geometry optimization is the distorted Fe<sub>4</sub> quadrilateral (not planar in some configurations) centered by carbon C<sub>2</sub> group (Table 1). The same geometrical feature was detected for some optimized structures of Fe<sub>12</sub>C<sub>12</sub> cluster in [8].

The energy gaps between occupied and vacant molecular states defined as the difference of energy of the highest occupied (HOMO) and the lowest unoccupied (LUMO) molecular orbitals for all investigated ground isomers are also presented in Table 1. In the majority of compositions, both functionals give the gap values of the same order of magnitude; however, for (6,5), (6,7), and (7,6) clusters, the

gaps obtained with PBE and BLYP potentials are considerably different. In some compositions, the HOMO and LUMO correspond to spin down orbitals, in the rest cases one of the orbital belongs to spin up and the other to spin down type. We do not found any ground isomer where both orbitals correspond to the spin up states. Typically, the HOMO and LUMO in investigated clusters contain 70–90% of the Fe 3d atomic orbital with admixtures of Fe4s, 4p and C2p character. In the small particles FeC<sub>4</sub> and Fe<sub>3</sub>C<sub>2</sub>, the HOMO and LUMO also contain C2s character (up to 12%). In the clusters containing greater number of atoms, the contributions of C2s orbitals in the HOMO and LUMO do not exceed 1%.

The absolute value of binding energy for Fe<sub>n</sub>C<sub>m</sub> gradually increases with increasing number of atoms; however, to evaluate the relative stability of nanoparticles with various sizes, it is more reasonable to compare the ground isomer binding energy per atom,  $E_{nm} = |E_{gs}|/(m + n)$ . The values of this specific energy (measured in eV/atom) for the investigated clusters are shown in Fig. 2. The results of ‘BLYP’ calculations are used in this figure; however, the trends of  $E_{nm}$  variation obtained with the use of ‘PBE’ functional are very similar to those shown in Fig. 2. For the presentation of  $E_{nm}$  values, we use the same style as of the map of observed Fe<sub>n</sub>C<sub>m</sub><sup>+</sup> compositions from Ref. [7]; in bold, we indicate  $E_{nm}$  for the experimentally observed (5,6), (6,6), (5,7), (6,7), (7,7), (5,8), (6,8), and (7,8) compositions. As can be seen, the value of  $E_{nm}$  gradually increases with increasing number of carbon atoms ‘m’ (number of Fe atoms ‘n’ is constant) by 1.5–0.4 eV on each step for the small particles and by 0.2–0.1 eV for the particles with  $n + m > 10$ . On the other hand, the specific energy decreases with increasing number of iron atoms ‘n’ (‘m’ is constant) by 0.2–0.1 eV on each step for all investigated clusters with  $n + m > 3$ . According to these trends, the most stable particle among investigated in our work is Fe<sub>4</sub>C<sub>8</sub>; however, this composition was not detected in experiment



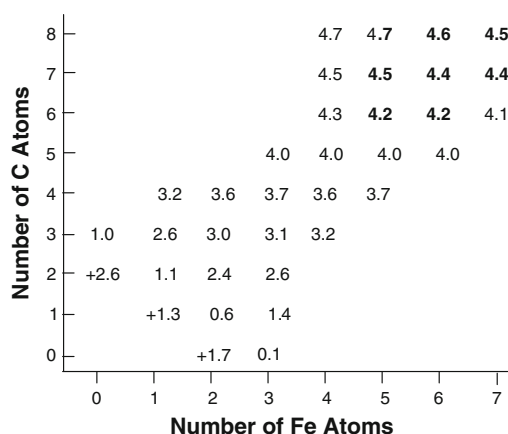
**Fig. 2** The absolute values of binding energy per atom (eV/atom) for the ground isomers of neutral Fe<sub>n</sub>C<sub>m</sub> nanoparticles

by Pilgrim and Duncan [7]; moreover, the greater abundance of Fe<sub>7</sub>C<sub>8</sub><sup>+</sup> [3] in comparison with (5,8) and (6,8) compositions is not in agreement with  $E_{nm}$  variation.

The absence of correlation between stability of neutral Fe<sub>n</sub>C<sub>m</sub> ground isomers and their experimental relative abundance, we could try to explain by the fact that results of Pilgrim and Duncan [7] correspond to the ionic Fe<sub>n</sub>C<sub>m</sub><sup>+</sup> clusters. To verify this suggestion, we considered the configurations and stability of charged Fe<sub>n</sub>C<sub>m</sub><sup>+</sup> particles. The search for ground isomers of Fe<sub>n</sub>C<sub>m</sub><sup>+</sup> was undertaken using the results obtained for neutral Fe<sub>n</sub>C<sub>m</sub> clusters. The geometry optimization of initial structures for each (n, m) composition of neutral particles usually leads to several stable geometries, of which 5–10 have the binding energy close to that of the ground isomer. These ‘more stable’ neutral configurations (as well as the ground state structures) were used for the investigation of ionization process and to obtain the cationic ground isomers. Firstly, we performed the calculations of all obtained Fe<sub>n</sub>C<sub>m</sub> particles with fixed atomic positions and with one electron removed from the cluster. On the second step, the ‘neutral’ configurations were subjected to geometry optimization with one electron removed from the cluster. The ionization energy of a cluster is defined as  $E_i = E_b^+ - E_b$ , where  $E_b^+$  and  $E_b$  are the binding energies of the charged and neutral particle, respectively. The ionization energies obtained on the first step ( $E_{i1}$ ) include the effects of electron density relaxation which take place during ionization process, while the ionization energies obtained on the second step ( $E_{i2}$ ) include the effects of electron density relaxation as well as the effects of relaxation of atomic positions. The ionization energies  $E_{i1}$  and  $E_{i2}$  of all ground isomers are summarized in Table 1. The analysis of ionization energies shows that in the majority of compositions,  $E_{i1}$  and  $E_{i2}$  are very close (the difference is less than 0.2 eV). However, in the cases of Fe<sub>2</sub>C<sub>4</sub>, Fe<sub>3</sub>C<sub>3</sub>(#2), Fe<sub>3</sub>C<sub>4</sub>, Fe<sub>7</sub>C<sub>6</sub>(#2), and Fe<sub>7</sub>C<sub>8</sub> (calculations using BLYP functional), this difference is noticeably greater (from 0.3 to 1 eV). Mainly, this effect is due to the noticeable transformation of geometrical structure during ionization of the cluster; however, in the cases of Fe<sub>3</sub>C<sub>3</sub>(#2) and Fe<sub>7</sub>C<sub>8</sub>, the reordering of spin moments on metal atoms also takes place. The latter effects will be considered in next section.

The energetic ordering of ionic structures usually was not the same as for the parent neutral configurations, after analysis of the results obtained in these calculations the lowest-energy geometry was then considered as the ground state for Fe<sub>n</sub>C<sub>m</sub><sup>+</sup> composition. The values of specific energy  $E_{nm}^+$  for the ionic particles are shown in Fig. 3, as in the previous case the results of ‘BLYP’ calculations were presented. As can be seen from Fig. 3, the values of  $E_{nm}^+$  considerably differ from those obtained for the corresponding neutral particles, especially for the compositions





**Fig. 3** The specific  $E_{nm}^+$  energy (eV/atom) for the ground isomers of charged  $\text{Fe}_n\text{C}_m^+$  nanoparticles ('+' of the values for dimers means that their binding energies are positive)

with  $n + m < 8$ . For  $\text{Fe}_2^+$ ,  $\text{FeC}^+$ , and  $\text{C}_2^+$  dimers, we obtained positive binding energies, this result means that such ionic species are not stable. The rest part of investigated charged clusters has negative binding energy; however,  $E_{nm}^+$  for the particles containing 3 atoms are several times less than corresponding  $E_{nm}$  for the neutral clusters. For the ionic particles containing 4 atoms, these energies are almost two times less than corresponding 'neutral' values, and for the 5–6-atom objects,  $E_{nm}^+$  are  $\sim 1.5$  times less than those for the corresponding neutral clusters. For the ground isomers with  $n + m > 7$ , this energy decreases less considerably, but in all cases, the difference in specific energy for charged and neutral particles is not less than 0.4–0.5 eV/atom. The analysis of  $E_{nm}^+$  variation shows that these values also gradually increase with increasing number of carbon atoms 'm' ('n' is constant) by at least 0.3 eV on each step for the small particles and by 0.2 eV for the particles with  $n + m > 10$ . However, the trends for 'horizontal' series (increasing 'n', 'm' is constant) for  $\text{Fe}_n\text{C}_m^+$  clusters are noticeably different for small clusters and for those with  $n + m > 7$ . For the former particles  $E_{nm}^+$  increases by 1.2–0.1 eV on each step, but for the latter compositions,  $E_{nm}^+$  variation is very smooth, e.g., the stability of ground isomers for  $\text{Fe}_n\text{C}_5^+$  ( $n = 3$ –6) is nearly the same. For  $\text{Fe}_n\text{C}_6^+$ ,  $\text{Fe}_n\text{C}_7^+$ , and  $\text{Fe}_n\text{C}_8^+$  series,  $E_{nm}^+$  slowly decreases by 0.2 eV/atom when  $n$  increases from 4 to 8. According to these results, we can suggest that the greater abundance of  $\text{Fe}_7\text{C}_8^+$  in comparison with (5,8) and (6,8) compounds is not directly connect with the stability of corresponding ground isomers.

On the other hand, according to the experimental conditions, the species with high ionization energies collect electrons from species with low ionization energies. Consequently, the cations extracted into the mass-spectrometer are likely to be those with the lowest ionization energies. It

is important to note that the values of  $E_i$  presented in Table 1 correspond to the isomers which are the ground configurations for the neutral clusters. However, as mentioned above, for each composition, the ionization process was also investigated for several 'more stable' isomers. The results of these calculations show that the ionization energies of some of the isomers are noticeably less than those presented in Table 1. For example, the lowest ionization energies for the (5,6), (5,7), (6,6), (5,8), (6,7), (6,8), (7,7), and (7,8) compositions (detected in experiment) are 6.1, 5.8, 5.8, 6.3, 6.2, 5.6, 5.8, and 6.2 eV, respectively. Figure 3 shows that the lowest  $E_{nm}^+$  value for the clusters observed in experiment (4.2 eV/atom) was obtained for the (5,6) and (6,6) compositions and their corresponding ionization energies are 6.1 and 5.8 eV. On the other hand, there are three more stable compositions not observed in experiment (Fig. 3):  $\text{Fe}_4\text{C}_6$  ( $E_{nm}^+ = 4.3$  eV/atom),  $\text{Fe}_4\text{C}_7$  ( $E_{nm}^+ = 4.5$  eV/atom) and  $\text{Fe}_4\text{C}_8$  ( $E_{nm}^+ = 4.7$  eV/atom). According to our calculations, the lowest ionization energies obtained for these compositions are: 6.2, 6.3, and 6.7 eV, respectively. Thus, the comparison of specific and ionization energies shows that both factors can be due to the experimental cation distribution. However, this analysis cannot explain the great experimental abundance of  $\text{Fe}_7\text{C}_8$  composition with  $E_{nm}^+ = 4.5$  eV/atom and the lowest value of ionization energy  $E_i = 6.2$  eV.

Another reason which could be responsible for different relative abundance is the number of stable conformations for each ( $n, m$ ) composition. It is evident that all possible isomers could be found for small clusters only, for this reason in the present work, we could evaluate only the number of 'more stable' configurations for each ( $n, m$ ) composition. For the most stable  $\text{Fe}_4\text{C}_8^+$  composition (not observed in experiment), we obtained 15 isomers, only 3 of them had binding energies in  $[E_{bg}; E_{bg} - 1 \text{ eV}]$  interval (where  $E_{bg}$  is the energy of ground structure) and no one of them fell into  $[E_{bg}; E_{bg} - 0.5 \text{ eV}]$  range. For further comparison, we marked this result as '0/3'. For  $\text{Fe}_5\text{C}_8^+$  (observed in experiment with small abundance), we obtained 5 isomers in '1 eV—limit' and 2 isomers belonging to '0.5 eV—limit'; thus, this composition can be characterized by '2/5' value. For  $\text{Fe}_6\text{C}_8^+$  (also observed in experiment with small abundance), the corresponding value is also 2/5. However, for the first composition with essential abundance  $\text{Fe}_7\text{C}_8^+$ , we obtained 11 isomers in  $[E_{bg}; E_{bg} - 1 \text{ eV}]$  interval and 7 structures belonging to  $[E_{bg}; E_{bg} - 0.5 \text{ eV}]$  interval, i.e., the 'density of isomers' for this composition can be defined as 7/11. For the less stable compositions with seven carbon atoms, we obtained 3/6 for  $\text{Fe}_4\text{C}_7^+$  (not observed in experiment), 4/6 for  $\text{Fe}_5\text{C}_7^+$  (observed in experiment), 7/7 for  $\text{Fe}_6\text{C}_7^+$ , and 4/4 for  $\text{Fe}_7\text{C}_7^+$  (both observed in experiment). For the clusters with six carbon atoms, these values are: 1/1 for  $\text{Fe}_4\text{C}_6^+$  (not

observed in experiment), 6/6 and 7/9 for  $\text{Fe}_5\text{C}_6^+$  and  $\text{Fe}_6\text{C}_6^+$  (both observed in experiment) and 3/3 for  $\text{Fe}_7\text{C}_6^+$  (not observed in experiment). It is evident that ‘1 eV’ and ‘0.5 eV’ limits used in such crude evaluation are quite arbitrary, and defined numbers of isomers are of qualitative nature. However, the analysis of stability of ground configurations and the numbers of most stable isomers allows one to conclude that the latter factor is also due to the experimental results obtained by Pilgrim and Duncan [7].

#### 4 Effective charges and total spin densities on atoms

The most popular analyses of charge and spin density distribution based on the Mulliken or related methods depend strongly on the basis set used in the calculation and on the method of partitioning the overlap populations of each pair of atoms to the individual atoms in the pair. The more ‘physical’ types of population analysis use spatial integration of electron density. One of such approach was suggested by Hirshfeld [24]; in this scheme, the partition of electron density between atoms is made by the use of a ‘partition function’,  $g_\alpha(\mathbf{r} - \mathbf{R}_\alpha)$  ( $\alpha$  is an atom index), which is large for small  $\mathbf{r} - \mathbf{R}_\alpha$  and small for large  $\mathbf{r} - \mathbf{R}_\alpha$ . The most popular choice for this function is  $g_\alpha(\mathbf{r}) = \rho_\alpha(\mathbf{r})$ , where  $\mathbf{r} = |\mathbf{r}_i - \mathbf{R}_\alpha|$  and  $\rho_\alpha(\mathbf{r})$  is the charge density for atom  $\alpha$ . The drawback of this method is the dependence of  $Q_{\text{eff}}$  on the shape of the partition function. On the other hand, the most natural boundary for electron density partitioning is the locus of points of the  $\rho(\mathbf{r})$  minimum. A method for the realization of this idea using numerical spatial integration incorporated in DV method was reported in Ref. [27]. Another advantage of this approach is the ability to separately integrate the spin up and spin down densities on a given atom. The sum of these values gives the effective charge ( $Q_{\text{eff}}$ ) of the atom, while their difference gives the total spin density ( $S_{\text{eff}}$ ) on the atom. In the previous work [17], this method was applied to the small  $\text{Fe}_n\text{C}_m$  nanoparticles, and the comparison of various schemes for the atomic effective charge calculations demonstrated that the most realistic values were obtained using the spatial numerical integration method [27]. However, in contrast to  $Q_{\text{eff}}$ , the magnetic moments on the atoms and their ordering in the small  $\text{Fe}_n\text{C}_m$  clusters were very similar when calculated using either DMol<sup>3</sup> (Hirshfeld) or DV (integration) methods [17].

The values of  $Q_{\text{eff}}$  in unit of  $e$ , where  $e$  is the electron charge, and  $S_{\text{eff}}$  in  $\mu_B$  obtained for the ground isomers of all investigated  $\text{Fe}_n\text{C}_m$  ground isomers are listed in Table 2. The order of  $Q_{\text{eff}}$  and  $S_{\text{eff}}$  values corresponds to the order of atomic coordinates in Appendix; this allows one to associate of  $Q_{\text{eff}}$  and  $S_{\text{eff}}$  results with specific atoms in the structures. As can be seen, the values of  $Q_{\text{eff}}$  computed by

the DV integration scheme [27] are 2–3 times greater than those calculated by Hirshfeld procedure [24]. In the former approach, the greatest charge on the Fe atom was found to be 1.08 in the  $\text{Fe}_6\text{C}_8$  (#1) cluster, while in the latter approach, the greatest charge on metal atom (0.4) was obtained in  $\text{FeC}_4$ . Similar results were obtained for carbon atoms, the greatest absolute value of  $|Q_{\text{eff}}|$  (DMol) on C atom was found to be 0.24 in the  $\text{Fe}_3\text{C}_3$  (#2) and  $\text{Fe}_4\text{C}_3$  clusters, the corresponding  $|Q_{\text{eff}}|$  (DVM) was found to be 1.21 for the of  $\text{Fe}_7\text{C}_7$  (#1) particle (Table 2).

As mentioned above, for the small  $\text{Fe}_n\text{C}_m$  clusters, the values of  $S_{\text{eff}}$  obtained in both approaches were very similar [17], and the same results were obtained in the present calculations. For the majority of  $(n, m)$  compositions, the values of  $S_{\text{eff}}$  on Fe and C atoms computed by DMol<sup>3</sup> and DVM are in surprisingly close agreement (Table 2). The considerable disagreement was obtained only for the ground isomer of  $\text{Fe}_3\text{C}_4$  cluster, where in the DMol calculations,  $S_{\text{eff}}$  of the two equivalent Fe atoms (2.70) is essentially greater than the total spin of these atoms obtained in DV approach (0.76). The comparison of electronic structure features of this cluster obtained in DMol and DV methods shows that the structure of Fe3d levels is considerably different in two approaches; in particular, the gap between HOMO (Fe3d spin down) and LUMO (Fe3d spin up) was found to be  $\sim 0.7$  eV in DMol calculations (Table 1) and only 0.2 eV in DVM calculations. Since the DMol<sup>3</sup>–GGA method provides a more accurate solution of the DFT equations than DVM–LDA, we can predict that all Fe atoms in the ground isomer of  $\text{Fe}_3\text{C}_4$  cluster will have the high spin configuration and their total spin density will be close to 2.7.

Table 2 shows the ferromagnetic ordering (FM) of spins in the ground isomers of almost all investigated compositions. However, in some cases, the antiferromagnetic (AFM) ordering (or the ordering were the spin of at least one metal atom has an opposite direction to those of the other Fe atoms) was obtained for one of the two possible ground isomers. For  $\text{Fe}_3\text{C}_3$  (#1),  $\text{Fe}_5\text{C}_6$  (#2),  $\text{Fe}_6\text{C}_5$  (#2),  $\text{Fe}_5\text{C}_7$  (#1), and  $\text{Fe}_6\text{C}_8$  (#2) particles, the total spin densities at all metal sites have the same direction, while the second ground configurations of these compositions demonstrate the spin ordering where some magnetic moments on Fe atoms have the opposite direction. There is only one (7,6) composition for which in both possible ground conformations two opposite spin directions on metal atoms were obtained. Moreover, in the case of  $\text{Fe}_7\text{C}_6$ , we did not find isomers with FM ordering neither for the neutral nor for the charged particles, so we can conclude that FM ordering for this composition is not probable at least for the more stable isomers. Also noteworthy are the results for the greatest and smallest values of  $S_{\text{Fe}}$ , the former were found to be 3.35 (DMol) and 3.41 (DVM) in the  $\text{Fe}_4\text{C}_5$  and  $\text{Fe}_4\text{C}_6$



**Table 2** Effective charges (e) and total spin densities ( $\mu_B$ ) on atoms in  $\text{Fe}_n\text{C}_m$  nanoparticles

Composition		Fe		C				
$\text{FeC}_4$	DMol	$Q_{\text{eff}}$	0.40		-0.03	-0.03	-0.17	-0.17
		$S_{\text{eff}}$	2.27		0.04	0.04	-0.02	-0.02
	DVM	$Q_{\text{eff}}$	0.94		-0.07	-0.07	-0.40	-0.40
		$S_{\text{eff}}$	3.17		0.06	0.06	0.22	0.22
$\text{Fe}_3\text{C}_2$ (#1)	DMol	$Q_{\text{eff}}$	0.12	0.11	-0.17	-0.17		
		$S_{\text{eff}}$	3.01	2.51	0.05	0.05		
	DVM	$Q_{\text{eff}}$	0.16	0.44	-0.52	-0.52		
		$S_{\text{eff}}$	3.16	3.19	0.06	0.06		
$\text{Fe}_3\text{C}_2$ (#2)	DMol	$Q_{\text{eff}}$	0.12	0.14	-0.23	-0.18		
		$S_{\text{eff}}$	2.69	2.80	-0.01	0.03		
	DVM	$Q_{\text{eff}}$	0.33	0.34	-0.48	-0.53		
		$S_{\text{eff}}$	2.91	2.90	-0.05	0.04		
$\text{Fe}_2\text{C}_4$	DMol	$Q_{\text{eff}}$	0.25	0.22	-0.02	-0.20	-0.07	-0.18
		$S_{\text{eff}}$	2.82	2.93	0.07	0.07	0.06	0.14
	DVM	$Q_{\text{eff}}$	0.47	0.51	-0.19	-0.28	-0.11	-0.40
		$S_{\text{eff}}$	3.31	3.28	0.08	0.14	0.06	0.10
$\text{Fe}_3\text{C}_3$ (#1)	DMol	$Q_{\text{eff}}$	0.12	0.20	-0.12	-0.18	-0.07	
		$S_{\text{eff}}$	2.16	2.99	-0.13	-0.15	0.03	
	DVM	$Q_{\text{eff}}$	0.43	0.42	-0.55	-0.34	-0.02	
		$S_{\text{eff}}$	2.38	2.99	-0.17	-0.18	0.04	
$\text{Fe}_3\text{C}_3$ (#2)	DMol	$Q_{\text{eff}}$	0.18	0.26	-0.24	-0.19	-0.18	
		$S_{\text{eff}}$	2.44	-1.67	0.02	0.10	0.01	
	DVM	$Q_{\text{eff}}$	0.76	0.76	-1.01	-0.47	-0.54	
		$S_{\text{eff}}$	2.47	-2.02	0.02	0.13	-0.06	
$\text{Fe}_3\text{C}_4$	DMol	$Q_{\text{eff}}$	0.22	0.25	-0.18	-0.18	-0.18	-0.18
		$S_{\text{eff}}$	2.65	2.70	0.02	0.02	0.02	0.02
	DVM	$Q_{\text{eff}}$	0.74	0.93	-0.65	-0.65	-0.65	-0.65
		$S_{\text{eff}}$	2.52	0.76	0.01	0.01	0.01	0.01
$\text{Fe}_4\text{C}_3$	DMol	$Q_{\text{eff}}$	0.16	0.20	-0.24	-0.23	-0.16	
		$S_{\text{eff}}$	2.58	2.88	-0.17	0.07	0.10	
	DVM	$Q_{\text{eff}}$	0.60	0.65	-0.88	-0.43	-0.53	
		$S_{\text{eff}}$	2.53	2.90	-0.25	0.10	0.12	
$\text{Fe}_3\text{C}_5$	DMol	$Q_{\text{eff}}$	0.16	0.28	-0.18	-0.18	-0.06	-0.15
		$S_{\text{eff}}$	2.14	2.95	0.11	0.11	0.11	-0.19
	DVM	$Q_{\text{eff}}$	0.67	0.80	-0.55	-0.55	-0.05	-0.56
		$S_{\text{eff}}$	2.38	3.04	0.07	0.07	0.11	-0.15

Table 2 continued

Composition			Fe				C																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																																							

Table 2 continued

Composition		Fe										C									
		$Q_{\text{eff}}$	0.17	0.13	0.19	0.18	0.11	0.12													
Fe <sub>6</sub> C <sub>5</sub> (#2)	DMol	$S_{\text{eff}}$	2.24	2.67	2.40	2.94	3.25	3.19													
	DVM	$Q_{\text{eff}}$	0.82	0.35	1.04	0.41	0.17	0.41													
Fe <sub>4</sub> C <sub>8</sub>		$S_{\text{eff}}$	2.52	2.68	2.34	2.96	3.24	3.26													
	DMol	$Q_{\text{eff}}$	0.24	0.29	0.29	0.29															
Fe <sub>5</sub> C <sub>7</sub> (#1)		$S_{\text{eff}}$	2.96	3.21	2.66	2.83															
	DVM	$Q_{\text{eff}}$	0.63	0.96	1.00	0.96															
Fe <sub>5</sub> C <sub>7</sub> (#2)		$S_{\text{eff}}$	3.15	3.40	3.20	3.16															
	DMol	$Q_{\text{eff}}$	0.23	0.23	0.21	0.23	0.24														
Fe <sub>6</sub> C <sub>6</sub>		$S_{\text{eff}}$	2.38	2.38	2.53	3.13	3.00														
	DVM	$Q_{\text{eff}}$	1.01	1.01	0.77	0.68	0.87														
Fe <sub>5</sub> C <sub>8</sub>		$S_{\text{eff}}$	2.33	2.33	2.50	3.09	3.09														
	DMol	$Q_{\text{eff}}$	0.21	0.08	0.21	0.26	0.24														
Fe <sub>7</sub> C <sub>6</sub> (#1)		$S_{\text{eff}}$	2.84	1.94	2.91	-2.86	2.98														
	DVM	$Q_{\text{eff}}$	0.75	0.31	0.62	0.97	0.66														
Fe <sub>6</sub> C <sub>7</sub>		$S_{\text{eff}}$	2.69	1.70	2.83	-3.11	3.00														
	DMol	$Q_{\text{eff}}$	0.18	0.17	0.16	0.16	0.20														
Fe <sub>5</sub> C <sub>6</sub>		$S_{\text{eff}}$	2.93	2.94	2.92	2.92	2.88														
	DVM	$Q_{\text{eff}}$	0.75	0.36	0.47	0.50	0.72														
Fe <sub>7</sub> C <sub>6</sub> (#2)		$S_{\text{eff}}$	2.90	2.95	2.96	2.95	2.84														
	DMol	$Q_{\text{eff}}$	0.21	0.11	0.11	0.29	0.29														
Fe <sub>6</sub> C <sub>8</sub>		$S_{\text{eff}}$	2.54	2.40	2.40	3.12	3.12														
	DVM	$Q_{\text{eff}}$	0.79	0.28	0.28	1.01	1.01														
Fe <sub>7</sub> C <sub>7</sub>		$S_{\text{eff}}$	2.59	2.46	2.46	3.14	3.14														
	DMol	$Q_{\text{eff}}$	0.20	0.18	0.19	0.22	0.18														
Fe <sub>5</sub> C <sub>7</sub> (#1)		$S_{\text{eff}}$	2.97	2.89	2.97	2.50	2.02														
	DVM	$Q_{\text{eff}}$	0.67	0.77	0.46	0.81	1.06														
Fe <sub>7</sub> C <sub>6</sub> (#2)		$S_{\text{eff}}$	2.95	2.93	2.99	2.67	1.63														
	DMol	$Q_{\text{eff}}$	0.14	0.17	0.12	0.12	0.19														
Fe <sub>6</sub> C <sub>6</sub>		$S_{\text{eff}}$	2.87	2.76	2.78	2.78	-2.37														
	DVM	$Q_{\text{eff}}$	0.61	0.47	0.37	0.37	0.79														
Fe <sub>5</sub> C <sub>6</sub> (#1)		$S_{\text{eff}}$	2.55	2.66	2.56	2.56	-2.65														
	DMol	$Q_{\text{eff}}$	0.10	0.13	0.13	0.14	0.20														
Fe <sub>6</sub> C <sub>7</sub> (#2)		$S_{\text{eff}}$	3.07	2.57	2.76	2.50	-2.78														
	DVM	$Q_{\text{eff}}$	0.28	0.24	0.37	0.48	0.79														
Fe <sub>7</sub> C <sub>6</sub> (#3)		$S_{\text{eff}}$	3.00	2.40	2.71	2.55	-2.89														
	DVM	$Q_{\text{eff}}$																			

Table 2 continued

Composition		C																Fe	
Fe <sub>6</sub> C <sub>8</sub> (#1)	DMol																	Fe	
		Q <sub>eff</sub>	0.20	0.19	0.22	0.22	0.20	0.19	0.14	0.14	0.14	0.14	0.14	0.02	0.02	0.14	0.14	0.19	0.19
Fe <sub>6</sub> C <sub>8</sub> (#2)	DVM	S <sub>eff</sub>	2.02	1.74	2.26	-2.27	-2.03	-1.74	0.04	-0.02	0.02	0.02	0.02	0.02	0.02	-0.03	-0.04	0.26	-0.26
		Q <sub>eff</sub>	1.03	1.05	0.85	0.84	0.98	1.08	-0.67	-0.61	-0.63	-0.64	-0.61	-0.63	-0.64	-0.61	-0.63	-1.02	-1.02
		S <sub>eff</sub>	2.06	1.10	2.40	-2.41	-2.08	-1.11	0.01	-0.02	0.02	0.01	0.01	0.01	0.01	-0.01	-0.01	0.24	-0.24
		Q <sub>eff</sub>	0.15	0.13	0.09	0.27	0.24	0.22	-0.17	-0.18	-0.17	-0.16	-0.06	-0.06	-0.06	-0.04	-0.16	-0.16	-0.16
	DVM	S <sub>eff</sub>	2.65	2.65	2.61	3.07	2.98	3.04	0.06	0.08	0.13	0.05	0.06	0.06	0.06	0.06	0.02	0.06	0.06
		Q <sub>eff</sub>	0.64	0.14	0.03	1.03	0.98	0.33	-0.58	-0.45	-0.43	-0.52	-0.07	-0.08	-0.08	-0.08	-0.52	-0.52	-0.48
		S <sub>eff</sub>	2.42	2.53	2.28	3.16	3.13	3.15	0.04	0.06	0.08	0.04	0.05	0.05	0.05	0.05	0.00	0.00	0.06
		Q <sub>eff</sub>	0.16	0.20	0.19	0.16	0.19	0.16	-0.17	-0.16	-0.18	-0.17	-0.16	-0.17	-0.17	-0.16	-0.17	-0.20	-0.20
	DVM	S <sub>eff</sub>	2.84	2.94	2.91	2.67	2.44	2.50	0.05	0.07	0.07	0.11	0.09	0.10	0.09	0.10	-0.18	-0.18	-0.18
		Q <sub>eff</sub>	0.42	0.84	0.58	0.38	1.01	0.70	-0.56	-0.51	-0.66	-0.59	-0.54	-0.66	-0.54	-0.66	-1.21	-1.21	-1.21
		S <sub>eff</sub>	2.78	2.98	2.91	2.57	2.48	2.63	0.07	0.08	0.06	0.10	0.07	0.10	0.07	0.10	-0.16	-0.16	-0.16
		Q <sub>eff</sub>	0.14	0.17	0.18	0.19	0.18	0.18	-0.16	-0.16	-0.16	-0.17	-0.18	-0.16	-0.17	-0.18	-0.20	-0.20	-0.20
	DVM	S <sub>eff</sub>	2.47	2.92	2.64	2.80	2.91	2.96	0.10	0.09	0.04	0.06	0.12	0.10	0.12	0.10	-0.16	-0.16	-0.16
		Q <sub>eff</sub>	0.77	0.16	0.88	0.59	0.77	0.68	-0.63	-0.60	-0.51	-0.53	-0.61	-0.58	-0.61	-0.58	-1.18	-1.18	-1.18
		S <sub>eff</sub>	2.46	2.85	2.62	2.73	2.93	2.96	0.10	0.08	0.04	0.05	0.10	0.07	0.10	0.07	-0.16	-0.16	-0.16
		Q <sub>eff</sub>	0.17	0.21	0.17	0.17	0.23	0.17	-0.17	-0.17	-0.18	-0.17	-0.15	-0.17	-0.15	-0.17	-0.15	-0.15	-0.18
	DVM	S <sub>eff</sub>	2.94	3.02	2.94	2.97	3.10	2.95	0.09	0.09	0.12	0.14	0.12	0.12	0.14	0.12	0.12	0.12	0.14
		Q <sub>eff</sub>	0.59	0.86	0.47	0.01	1.02	0.50	-0.54	-0.55	-0.62	-0.54	-0.50	-0.58	-0.54	-0.50	-0.51	-0.51	-0.49
		S <sub>eff</sub>	2.86	3.02	2.84	2.95	2.98	2.84	0.09	0.09	0.08	0.13	0.09	0.08	0.13	0.09	0.08	0.09	0.12
		Q <sub>eff</sub>																	




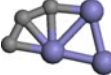
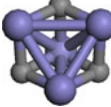
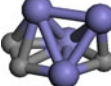
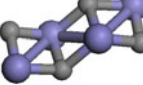
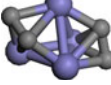
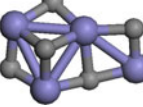
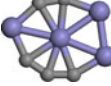
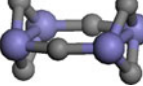
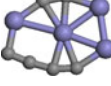
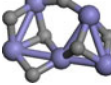
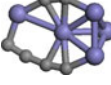
clusters, respectively (both have FM ordering), while the latter value 1.67 (DMol) and 1.10 (DVM) were obtained for the clusters  $\text{Fe}_3\text{C}_3$  and  $\text{Fe}_6\text{C}_8(\#1)$ , respectively (both have AFM ordering).

In contrast to the results obtained for the neutral  $\text{Fe}_n\text{C}_m$  clusters, FM ordering in the ground isomers of charged  $\text{Fe}_n\text{C}_m^+$  particles is not so preferable. Our calculations of two types of ionization show that the transformation of FM ordering to AFM one is not a simple process. In the cases of  $\text{Fe}_2\text{C}_4$ ,  $\text{Fe}_4\text{C}_8$ ,  $\text{Fe}_5\text{C}_7(\#1)$ ,  $\text{Fe}_6\text{C}_6$ ,  $\text{Fe}_5\text{C}_8$ ,  $\text{Fe}_6\text{C}_7$ ,  $\text{Fe}_7\text{C}_7(\#1)$ , and  $\text{Fe}_7\text{C}_8$ , the FM ordering transforms to the AFM type during simple ionization (without relaxation of atomic position) and the resulting ordering do not change during geometry optimization. For almost all compositions mentioned above, with the exception of  $\text{Fe}_7\text{C}_7(\#1)$ , one electron is removed from the HOMO corresponding to spin down states, but during self-consistency, the number of spin down orbitals in valence band increases. As a result, the spin moments on one or two metal atoms change their direction. In the case of  $\text{Fe}_4\text{C}_6$ , the FM ordering remains during simple ionization; however, the direction of spin moment on one Fe atom changes during relaxation of

atomic positions. It is interesting to note that in the cases of  $\text{Fe}_3\text{C}_3(\#2)$  and  $\text{Fe}_6\text{C}_5(\#1)$ , we obtained the transformation of initial AFM ordering to FM type during ‘simple’ ionization; however, the ionization with geometry optimization returned the spin moments on metal atoms to the initial orientations. In all other cases, the initial AFM ordering obtained for the neutral particles did not change for the corresponding ionic clusters. On the other hand, the values of  $S_{\text{Fe}}$  for the charged clusters were obtained in the same limits as for the neutral species; the greatest and smallest total spin densities at the metal atoms of  $\text{Fe}_n\text{C}_m^+$  were found to be 3.4 ( $\text{Fe}_5\text{C}_4^+$ , FM ordering) and 1.7 ( $\text{Fe}_6\text{C}_8^+$ , AFM ordering).

Finally, we compared the stabilities of  $\text{Fe}_n\text{C}_m$  nanoparticles generated by two ways. The construction of Fe–C clusters by addition of atoms ‘one by one’ used in the present work corresponds to experimental technology of laser vaporization of Fe metal and cooling in acetylene; however, Fe–C clusters can also be produced by grinding of solid-phase carbides. For the comparison of two types of clusters, we considered the small fragments of the real crystal lattice of the most stable solid-phase carbide  $\text{Fe}_3\text{C}$

**Table 3** Comparison of the values of binding energies (eV) of the fragments of  $\text{Fe}_3\text{C}$  crystal lattice and ground isomers of  $\text{Fe}_n\text{C}_m$  nanoparticles of the same composition

Composition	Fragments of $\text{Fe}_3\text{C}$ crystal lattice		Nanoparticles	
	Shape	$E_b$	Shape	$E_b$
$\text{Fe}_3\text{C}_2$		−18.5		−19.0
$\text{Fe}_3\text{C}_3$		−22.2		−25.2
$\text{Fe}_4\text{C}_3$		−28.1		−29.0
$\text{Fe}_4\text{C}_4$		−33.3		−35.6
$\text{Fe}_4\text{C}_5$		−39.1		−42.9
$\text{Fe}_4\text{C}_6$		−44.1		−49.6
$\text{Fe}_5\text{C}_6$		−48.1		−53.0



(cementite). It is evident that the stoichiometry of crystal fragments containing hundreds or thousands atoms will be close to (3,1); however, the small  $\text{Fe}_n\text{C}_m$  fragments of crystal lattice could be of various compositions. We considered seven types of crystal fragments with atomic positions corresponding to cementite lattice parameters reported in Ref. [28]. These initial structures were subjected to geometry optimization, leading to the new equilibrium geometries corresponding to the isolated clusters. In Table 3, we present the results of crystal fragments calculations, which are compared with the results obtained for corresponding  $\text{Fe}_n\text{C}_m$  clusters generated according to ‘binomial’ scheme. As can be seen, the absolute values of  $E_b$  of all optimized crystal fragments of either compact three-dimensional type or ‘chain-type’ are noticeably less than the energies obtained for corresponding nanoparticles. The comparison of these values shows that in general, the difference in stability increases with increasing number of atoms from 0.5 eV ( $\text{Fe}_3\text{C}_2$ ) to 5.5 eV ( $\text{Fe}_4\text{C}_6$ ).

## 5 Conclusions

Our investigations of  $\text{Fe}_n\text{C}_m$  nanoparticles containing 5–15 atoms do not confirm the earlier results [8] that the crystal

lattice fragments of iron carbides can be the better starting point to achieve the most stable geometry of isolated clusters. Moreover, we can suppose that the construction of binary  $\text{Me}_n\text{C}_m$  particles by addition of atoms ‘one by one’ (corresponding to vaporization–cooling technologies) leads to more stable objects for the wide range of metcars.

The analysis of spin moments ordering shows that in the majority of ground isomers of neutral  $\text{Fe}_n\text{C}_m$  clusters investigated in the present work, the FM ordering is preferable. On the other hand, our calculations of the similar Fe–O species which are in progress now show that FM ordering is not realized in almost all ground isomers and is rarely obtained for the more stable structures of  $\text{Fe}_n\text{O}_m$  nanoparticles. Therefore, we can conclude that the binary Fe–C clusters can serve as a perspective source to obtain the molecular objects with high magnetic moments.

**Acknowledgments** This work was supported by the Russian Foundation for Basic Research, grants 10-03-00152 and 09-03-00070.

## Appendix

See Table 4.

**Table 4** Atomic coordinates ( $\text{\AA}$ ) in the ground isomers of  $\text{Fe}_n\text{C}_m$  nanoparticles obtained using BLYP functional

Composition	Fe			C		
$\text{FeC}_4$	0.0	0.0	1.179	0.692	0.0	−0.689
				−0.692	0.0	−0.689
				1.719	0.0	0.100
				−1.719	0.0	0.100
$\text{Fe}_3\text{C}_2$ (#1)	0.0	0.0	−1.184	−0.649	0.0	0.769
	−2.213	0.0	−0.177	0.649	0.0	0.769
	2.213	0.0	−0.177			
$\text{Fe}_3\text{C}_2$ (#2)	1.528	0.645	0.0	−0.918	−0.983	0.0
	−0.513	0.609	1.156	0.415	−0.879	0.0
	−0.513	0.609	−1.156			
$\text{Fe}_2\text{C}_4$	−0.159	−1.306	0.0	0.014	0.770	0.0
	2.269	−0.403	0.0	−2.073	−0.618	0.0
				−1.312	0.423	0.0
				1.260	1.134	0.0
$\text{Fe}_3\text{C}_3$ (#1)	−0.164	−0.845	0.0	−0.135	1.149	0.0
	1.696	0.853	0.0	−2.087	−0.403	0.0
	2.106	−1.486	0.0	−1.415	0.732	0.0
$\text{Fe}_3\text{C}_3$ (#2)	−0.497	−1.338	−0.316	1.267	−1.227	−0.275
	1.317	0.504	−0.230	−0.388	1.394	−0.299
	−0.401	0.251	1.469	−1.298	0.417	−0.349
$\text{Fe}_3\text{C}_4$	0.0	−1.743	−0.386	1.530	0.658	−0.108
	0.0	−1.743	−0.386	1.530	−0.658	−0.108

**Table 4** continued

Composition	Fe			C		
	0.0	0.0	1.203	−1.530	−0.658	−0.108
				−1.530	0.658	−0.108
Fe <sub>4</sub> C <sub>3</sub>	−0.864	−0.957	−0.934	0.534	−1.762	−0.059
	0.899	−0.036	0.638	−0.180	1.657	0.272
	−1.352	0.238	1.306	−0.694	0.932	−0.685
	−0.843	−2.073	1.192			
Fe <sub>3</sub> C <sub>5</sub>	−0.006	0.858	0.0	1.813	−0.048	0.645
	0.420	−0.364	−2.001	1.813	−0.048	−0.645
	0.420	−0.364	2.001	−1.809	−0.070	0.0
				−1.325	0.019	1.250
				−1.325	0.019	−1.250
Fe <sub>4</sub> C <sub>4</sub>	0.939	−0.677	−1.389	1.334	−0.951	0.729
	−0.939	0.677	−1.389	−1.334	−0.951	0.729
	−0.897	−0.941	0.509	−0.788	1.994	0.151
	0.897	0.941	0.509	0.788	−1.994	0.151
Fe <sub>4</sub> C <sub>5</sub>	0.069	0.009	0.0	−0.994	1.851	0.0
	−2.414	0.565	0.0	0.279	2.069	0.0
	2.100	1.407	0.0	−0.558	−1.903	0.0
	2.407	−0.938	0.0	0.767	−1.914	0.0
				−1.656	−1.147	0.0
Fe <sub>5</sub> C <sub>4</sub>	0.0	0.0	−2.244	0.0	1.939	0.149
	0.0	−1.219	1.977	0.0	−1.939	0.149
	0.0	1.219	1.977	0.0	−1.556	−1.095
	1.153	0.0	0.091	0.0	1.556	−1.095
	−1.153	0.0	0.091			
Fe <sub>4</sub> C <sub>6</sub>	0.401	0.168	0.0	−0.754	2.968	0.0
	−2.296	0.816	0.0	0.503	2.244	0.0
	2.347	1.685	0.0	−0.108	−1.869	0.0
	2.749	−0.667	0.0	1.175	−1.782	0.0
				−2.604	−1.041	0.0
				−1.412	−1.523	0.0
Fe <sub>5</sub> C <sub>5</sub>	−0.099	0.131	0.0	−1.164	1.942	0.0
	−2.601	0.672	0.0	0.108	2.192	0.0
	1.918	1.441	0.0	−0.752	−1.779	0.0
	1.924	−0.840	1.145	0.607	−1.836	0.0
	1.924	−0.840	−1.145	−1.866	−1.083	0.0
Fe <sub>4</sub> C <sub>7</sub>	0.502	0.227	0.0	−0.950	1.819	0.0
	−2.468	0.637	0.0	0.229	2.329	0.0
	2.087	2.235	0.0	−0.453	−1.788	0.0
	2.573	−1.263	0.0	0.821	−1.887	0.0
				−3.005	−1.208	0.0
				−1.794	−1.607	0.0
				2.460	0.506	0.0
Fe <sub>5</sub> C <sub>6</sub> (#1)	−1.260	0.881	1.296	−2.753	0.358	0.0
	−1.260	0.881	−1.296	−2.084	−0.751	0.0
	0.825	1.186	0.0	0.470	1.179	1.867
	−0.119	−1.072	0.0	0.470	1.179	−1.867
	2.583	−1.018	0.0	1.565	−0.412	−1.492

**Table 4** continued

Composition	Fe			C		
				1.565	−0.412	−1.492
Fe <sub>5</sub> C <sub>6</sub> (#2)	0.219	0.042	−0.041	−0.945	1.822	−0.177
	−2.570	0.767	−0.175	0.307	2.118	−0.070
	2.145	1.552	0.173	−0.594	−1.838	0.617
	2.369	−0.889	0.740	0.676	−1.847	0.801
	2.344	−0.122	−1.434	−3.055	−1.058	0.171
				−1.897	−1.547	0.393
Fe <sub>6</sub> C <sub>5</sub> (#1)	1.191	1.703	0.0	0.324	−1.785	0.0
	−1.135	0.875	0.0	0.186	1.215	1.595
	1.367	−0.483	1.125	0.186	1.215	−1.595
	1.367	−0.483	−1.125	−0.700	0.301	−1.986
	−1.043	−1.430	−1.243	−0.700	0.301	1.986
	−1.043	−1.430	1.243			
Fe <sub>6</sub> C <sub>5</sub> (#2)	1.561	−0.446	−1.118	1.600	0.220	0.947
	−1.006	−0.606	−1.007	−1.593	0.745	0.714
	0.179	1.366	0.086	−2.240	−0.392	0.776
	−0.244	−0.714	1.468	1.771	−1.086	0.869
	0.569	−2.443	−0.083	0.276	0.683	−1.638
	−1.873	−2.328	0.490			
Fe <sub>4</sub> C <sub>8</sub>	0.563	0.234	0.0	−0.198	2.261	0.0
	−2.830	0.461	0.0	−0.724	−1.867	0.0
	1.571	2.844	0.0	0.541	−1.978	0.0
	2.400	−1.775	0.0	−3.300	−1.407	0.0
				−2.066	−1.726	0.0
				2.789	0.028	0.0
				2.558	1.300	0.0
				−1.305	1.626	0.0
Fe <sub>5</sub> C <sub>7</sub> (#1)	−1.193	0.869	1.225	−2.718	0.132	0.0
	−1.193	0.869	−1.225	−1.995	−0.938	0.0
	1.114	0.939	0.0	0.430	−0.044	1.910
	−0.036	−1.238	0.0	0.430	−0.044	−1.910
	2.558	−1.248	0.0	1.512	−0.654	−1.525
				1.512	−0.654	1.525
				−0.421	2.010	0.0
Fe <sub>5</sub> C <sub>7</sub> (#2)	−1.064	1.188	1.671	−2.277	0.041	0.773
	−0.597	−0.318	−0.252	−2.523	−0.880	−0.156
	0.935	1.552	0.016	0.808	1.362	2.089
	−0.483	−2.313	−1.965	1.137	0.215	−1.389
	1.789	−0.786	0.384	0.988	−1.056	−1.665
				1.803	0.608	1.732
				−2.016	−1.614	−1.139
Fe <sub>6</sub> C <sub>6</sub>	1.177	2.060	0.0	0.529	−1.930	0.0
	−0.956	0.961	0.0	0.353	1.559	1.695
	1.210	−0.356	1.159	0.353	1.559	−1.695
	1.210	−0.356	−1.159	−0.491	0.624	−2.064
	−1.105	−1.146	−1.428	−0.491	0.624	2.064
	−1.105	−1.146	1.428	−0.684	−2.452	0.0
Fe <sub>5</sub> C <sub>8</sub>	2.235	0.067	0.0	1.989	0.018	1.916

**Table 4** continued

Composition	Fe			C		
Fe <sub>6</sub> C <sub>7</sub>	0.116	0.189	1.196	1.989	0.018	−1.916
	0.116	0.189	−1.196	1.192	−0.061	−2.938
	−0.654	−0.179	−3.512	1.192	−0.061	2.938
	−0.654	−0.179	3.512	−1.963	0.026	0.678
				−1.963	0.026	−0.678
				−1.797	−0.027	−1.958
				−1.797	−0.027	1.958
	−1.251	1.261	1.170	−2.277	−0.053	−0.200
	−1.279	0.979	−1.458	−2.025	−0.607	0.936
	0.869	1.178	0.057	0.437	0.981	2.155
	−0.477	−1.344	−0.643	0.475	0.257	−1.858
	1.843	−1.121	0.509	1.464	−0.377	−1.268
	−0.374	−1.102	1.911	1.473	0.221	1.891
				0.475	−2.238	0.750
Fe <sub>7</sub> C <sub>6</sub> (#1)	0.801	2.077	0.0	−0.164	1.522	1.631
	−1.364	0.721	0.0	−0.164	1.522	−1.631
	1.104	−0.203	1.156	−0.803	0.443	−2.020
	1.104	−0.203	−1.156	−0.803	0.443	2.020
	−1.220	−1.405	−1.474	−0.648	−2.579	0.0
	−1.220	−1.405	1.474	0.423	−1.800	0.0
	2.952	0.866	0.0			
Fe <sub>7</sub> C <sub>6</sub> (#2)	1.853	1.642	−0.350	−0.229	1.073	2.536
	−0.415	1.097	0.362	0.362	1.426	−1.640
	1.142	−0.391	1.420	−0.706	0.685	−1.803
	0.914	−0.532	−1.053	−1.103	0.186	2.161
	−1.576	−1.033	−1.472	−0.310	−2.219	−0.606
	−1.345	−1.368	1.048	0.517	−2.011	0.411
	1.414	1.944	1.985			
Fe <sub>6</sub> C <sub>8</sub> (#1)	−1.169	1.429	1.219	−2.145	−0.001	−0.233
	−1.128	1.177	−1.378	−1.926	−0.543	0.936
	1.059	1.309	0.015	0.452	0.869	2.127
	−0.475	−1.354	−0.600	0.451	0.144	−1.779
	−0.351	−1.099	1.925	1.460	−0.418	−1.163
	1.853	−1.249	0.536	1.505	0.163	1.811
				0.431	−2.297	0.786
Fe <sub>6</sub> C <sub>8</sub> (#2)				−0.514	2.370	−0.203
	2.392	0.004	−0.074	1.960	0.088	1.830
	−0.021	−0.227	1.116	2.084	−0.050	−1.982
	0.259	0.603	−1.231	1.258	−0.075	−2.978
	−0.621	−0.189	−3.399	1.136	0.030	2.852
	−0.655	−0.015	3.505	−2.105	−0.043	0.802
	1.225	1.804	0.814	−1.810	0.075	−0.521
Fe <sub>7</sub> C <sub>7</sub> (#1)				−1.667	0.094	−1.813
				−1.936	−0.099	2.080
	−1.325	0.492	1.165	−2.443	0.117	−0.637
	−1.109	1.221	−1.491	−2.639	−0.960	0.079
	0.905	1.100	0.098	0.404	0.756	2.088
	−0.519	−1.055	−0.646	0.627	0.425	−1.876

**Table 4** continued

Composition	Fe			C		
Fe <sub>7</sub> C <sub>7</sub> (#2)	1.690	−1.338	0.396	1.488	−0.394	−1.323
	0.066	−1.298	2.354	1.522	0.077	1.839
	−1.755	−2.266	1.143	0.090	−2.376	0.811
	1.827	2.048	−0.181	−0.403	1.070	2.621
	−0.636	1.035	0.347	0.556	1.551	−1.489
	0.851	−0.601	1.915	−0.368	0.694	−1.836
	1.081	−0.412	−0.515	−1.273	0.182	2.219
	−1.171	−1.007	−1.521	−0.202	−2.245	−0.412
Fe <sub>7</sub> C <sub>8</sub>	−1.514	−1.250	0.937	0.427	−2.150	0.740
	1.251	1.905	2.172	2.089	0.680	1.004
	−1.422	0.586	1.242	−2.511	0.379	−0.630
	−1.109	1.236	−1.612	−2.851	−0.622	0.130
	0.801	0.997	0.106	0.348	0.658	2.154
	−0.587	−1.037	−0.498	0.632	0.439	−1.933
	1.951	−1.296	0.320	1.541	−0.305	−1.372
	−0.042	−1.428	1.946	1.484	0.048	1.884
	−2.312	−2.244	0.989	0.668	−2.816	0.536
				−0.588	−3.093	0.738

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